

Introduction

- **Motivation:** Most MD simulations involving ferrite (the solid solution of carbon in BCC iron) reported in the literature are carried out on pure iron.
- **Probable reasons:** Influence of C in ferrite is neglected maybe due to its low carbon content (0.008 % in weight).
- **Carbon atoms** strengthen the Fe lattice by creating strain fields and restricting dislocation movement.

- **This work:** Tensile test simulations of ferrite and α -Fe were carried out on the [111] direction at 300, 500 and 700 K.
- **Fast strain rates:** Strain rates applied were high (10^7 – 10^{10} s⁻¹) due to MD time scale limitations.
- **The aim is to compare the tensile properties of ferrite and α -Fe at different strain rates and to estimate the influence of carbon atoms on the mechanical behaviour of ferrite.**

Building Geometry

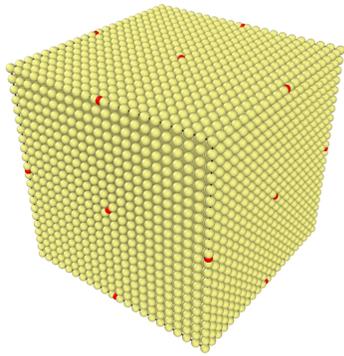


Fig. 1: Ferrite unit cell, Space group Im-3m. Carbon atoms (red) occupy octahedral sites in ferrite (Wyckoff position 6b [2]).

Simulation Set Up

Parameter	Set Up
Number of atoms	2.3×10^6
Dimensions	$300 \times 300 \times 300 \text{ \AA}^3$
Interatomic potential	Hepburn and Ackland [1]
Integration timestep	1 fs
Boundary conditions	ppp

Table 1: Parameters used in the simulations. p: periodic boundary conditions.

Equilibration

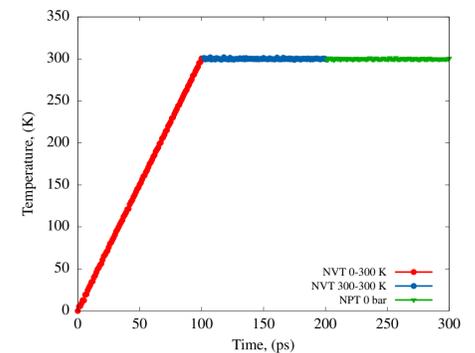


Fig. 2: Temperature versus time and ensembles used during equilibration.

Tensile Test Simulations

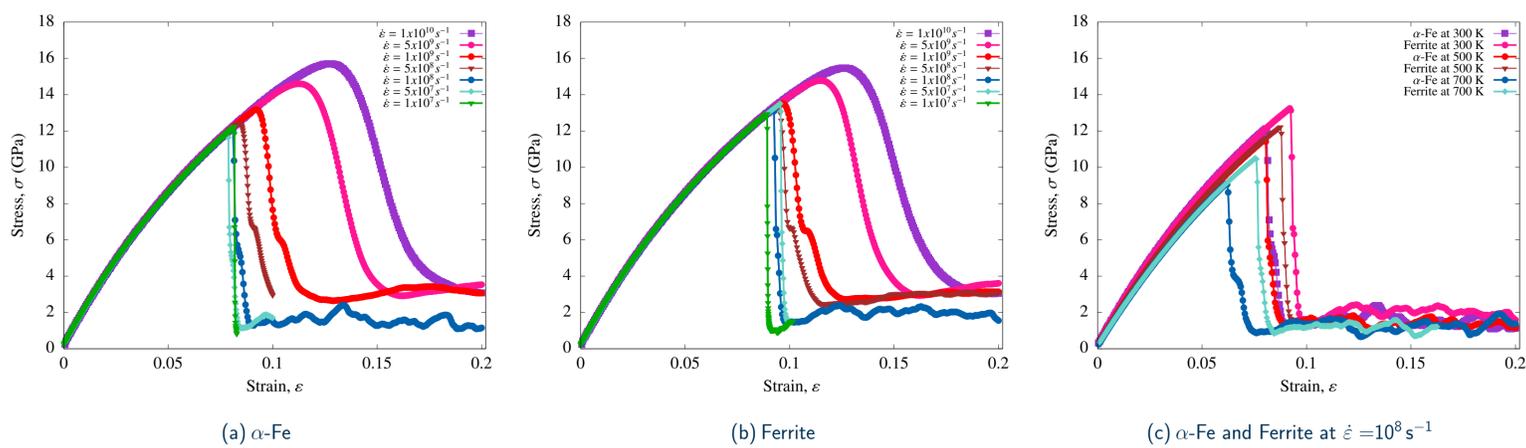


Fig. 3: Stress-strain curves showing that the difference in mechanical behaviour between α -Fe and ferrite become more significant at lower strain rates and higher temperatures.

Strain Rate Sensitivity

Table 2: The strain rate and yield strength relationship is described by $\sigma_y = C\dot{\epsilon}^m$.

	Temperature	m	C (GPa)	$\sigma_{\dot{\epsilon}=10^{-3}\text{s}^{-1}}$ (GPa)
α -Fe	300 K	0.035	6.86	5.40
	500 K	0.049	4.70	3.34
	700 K	0.069	2.70	1.67
Ferrite	300 K	0.022	9.34	8.02
	500 K	0.030	7.42	6.01
	700 K	0.037	5.69	4.39

Deformation Mechanism

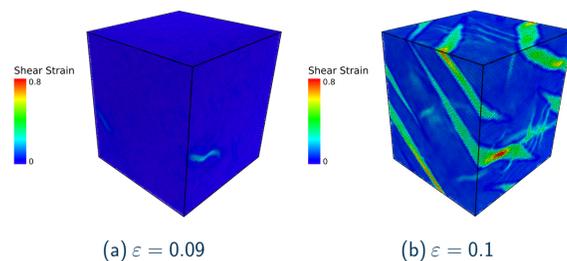


Fig. 4: Deformation process of Ferrite at 700 K. (a) Dislocation nucleation. (b) Dislocation propagation and activation of secondary slip planes.

Stresses and Slip Planes

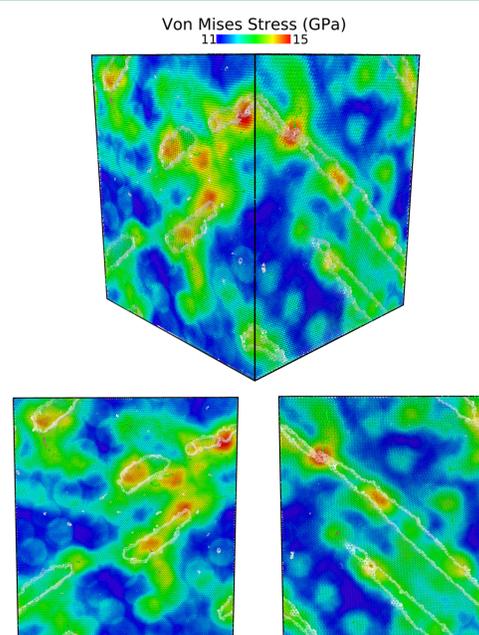


Fig. 5: Typical stress map after yield. Regions of higher stress are located along slip planes and dislocation junctions. The stress field acting on C atoms can also be visualised.

Carbon Diffusion

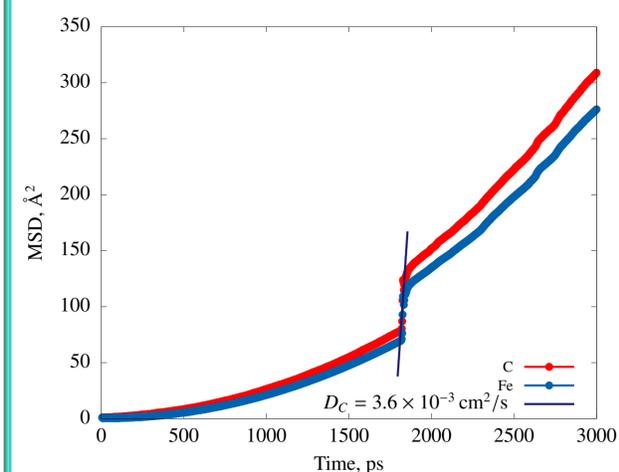


Fig. 6: Mean square displacement of C and Fe versus time. During the deformation process, three stages can be identified: an elastically dominated region, yielding and a plastically dominated region.

Conclusions

- An important difference between the mechanical behavior of pure iron and ferrite indicates that using only Fe atoms to represent ferrite will result in errors.
- The strain rate sensitivity of the yield strength is described by $\sigma_y = C\dot{\epsilon}^m$, with higher m values for α -Fe than for ferrite.
- The value D_C during yield was calculated as $3.6 \times 10^{-3} \text{ cm}^2/\text{s}$.

References

- [1] Hepburn, D. J. and Ackland, G. J. *Phys. Rev. B*. 2008, 78, 165115
- [2] Bhadeshia, H. K. D. H. *Journal of Materials Science*. 2004, 39, 3949–3955

Acknowledgements

- This research was funded by the Science and Technology Council of Mexico.
- This research made use of the Topsy High Performance Computing service at Newcastle University.